Controlling H⁺ vs CO₂ Reduction Selectivity on Pb Electrodes
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Supporting Information

ABSTRACT: Nanocrystalline Pb films prepared by reducing PbO₂ precursors have up to 700-fold lower H⁺ reduction activity than polycrystalline Pb foil electrodes but maintain the ability to reduce CO₂. As a result, these “oxide-derived” Pb (OD–Pb) electrodes have higher Faradaic efficiency for CO₂ reduction to HCO₃⁻ in aqueous solutions with almost no competitive H⁺ reduction. Even with very low CO₂ concentrations in N₂-saturated NaHCO₃ solution, OD–Pb converts CO₂ derived from HCO₃⁻ decomposition to HCO₃⁻ with almost quantitative Faradaic efficiency while Pb foil has less than 10% efficiency. Electrokinetic data suggest that the difference in selectivity between the two electrodes is caused by a difference in the coverage of a surface layer—likely a metastable Pb oxide—that is passivating for H⁺ reduction but active for CO₂ reduction.

KEYWORDS: energy, CO₂, electroreduction, fuel, formate and Pb

INTRODUCTION

The use of renewable electricity to power the conversion of CO₂ and H₂O into valuable chemicals could reduce net CO₂ emissions.¹⁻⁵ The success of this strategy hinges on the development of efficient electroreduction catalysts that selectively and efficiently reduce CO₂ at high rates using H₂O as a H⁺ source. One of the fundamental challenges in catalyst development is to suppress H⁺ reduction to H₂ without compromising CO₂ reduction. Most materials have a strong preference for H⁺ reduction over CO₂ reduction in aqueous electrolytes unless extreme overpotentials are applied, which compromises energetic efficiency.⁶

We have recently explored the use of metal oxides as catalyst precursors and metastable catalytic species to address the challenge of CO₂ versus H⁺ reduction selectivity in aqueous electrolytes. In the case of Sn electrodes, we showed that CO₂ reduction requires metastable surface Sn oxides and that enhancing the oxide content of electrodes improves selectivity.⁷ For Au and Cu electrodes, oxide layers are not stable under CO₂ reduction conditions. However, Au and Cu electrodes prepared by reducing Au oxide and Cu oxide precursors—“oxide-derived” metals—have very different catalytic properties than bulk materials or nanoparticles.⁸⁻¹¹ Oxide-derived Cu (OD–Cu) and oxide-derived Au (OD–Au) are composed of thin films of interconnected nanocrystallites with 10⁻100 nm dimensions. In CO₂ reductions, these electrodes have higher selectivity for CO₂ reduction versus H₂ evolution at low overpotential compared to their bulk or nanoparticle counterparts. For OD–Au, this difference is the result of its higher specific (i.e., surface-area-normalized) activity for CO₂ reduction.⁹ For OD–Cu, the difference is primarily the result of its lower specific H⁺ reduction activity.⁶ These studies demonstrate that the microstructure and morphology that result from metal oxide reduction alter the catalytic properties of these metals.

This study examines the factors that control the selectivity for H⁺ versus CO₂ reduction on Pb electrodes by comparing Pb foil to oxide-derived Pb (OD–Pb). Several previous studies have evaluated the CO₂ reduction properties of Pb using a variety of different electrode structures, electrochemical cells, and electrolytes.⁶ In CO₂-saturated aqueous HCO₃⁻, Pb forms primarily HCO₃⁻ and H₂. The equilibrium potential for the CO₂/HCO₃⁻ couple in this electrolyte is approximately 0 V versus the reversible hydrogen electrode (RHE; all potentials are reported with respect to this reference).¹² All studies have shown that appreciable CO₂ reduction requires very large overpotentials, but reported results vary widely. Studies with Pb foil electrodes in simple electrochemical H-cells have reported 73%⁻⁹⁷% Faraday efficiency (FE) for HCO₃⁻ at -1.2 V,⁶,¹³ 10% FE at -1.3 V,¹⁴ and 20%⁻⁵⁰% at -1.1 to -1.9 V.¹⁵

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Although some variability is caused by differences in the CO₂ flow rate and HCO₃⁻ concentration, these results suggest that the reaction is sensitive to subtle features of the Pb surface structure. Interestingly, switching from Pb foil to an electrodeposited Pb electrode improved the FE at −1.1 V from 20% to >90%, although the FE declined over time on the electrodeposited electrode.¹⁵ Experiments with Pb electrodes in gas diffusion cells and liquid flow cells have yielded a range of FEs and current densities that are comparable to the range obtained with Pb foil in H-cells.¹⁶⁻²³

Here we show that OD−Pb has greatly suppressed H⁺ reduction activity compared to polycrystalline Pb foil. This feature leads to improved FE for CO₂ reduction to HCO₃⁻, including almost quantitative FE in N₂-saturated HCO₃⁻ solution with very low CO₂ concentration. The Tafel behavior for H⁺ reduction and CO₂ reduction on Pb foil and OD−Pb suggest that the electrodes have different degrees of coverage of a surface layer that is passivating for H₂ evolution but active for CO₂ reduction. This layer is most likely a very thin, metastable Pb²⁺ oxide or hydride. The morphology and microstructure of OD−Pb promote a higher coverage of this layer relative to Pb foil.

**RESULTS**

OD−Pb electrodes were prepared by reducing thick Pb oxide films on polycrystalline Pb foil substrates. The Pb oxide films were grown by long pulsed anodizations in acidic solution (see Supporting Information).²⁴ X-ray diffraction and X-ray photoelectron spectroscopy (XPS) indicated that the oxide was primarily PbO₂ (Figure S1). The oxidized electrodes were used directly in H₂ evolution or CO₂ reduction electrolyses, during the initial period of which the PbO₂ film was reduced to OD−Pb. Figure 1 shows scanning electron microscopy (SEM), XPS, and grazing incidence X-ray diffraction (GIXD) for a representative OD−Pb electrode after the reduction of the PbO₂ precursor in CO₂-saturated 0.5 M NaHCO₃, the electrolyte used for CO₂ reduction (see below). SEM showed that the OD−Pb layer was ∼3 μm thick and composed of aggregated particles with 100–500 nm dimensions. Only peaks associated with Pb⁰ were observed by GIXD. Williamson–Hall analysis of the diffraction line shape indicated an average crystallite size of ∼45 nm (Figure S2). High-resolution XPS spectra showed peaks corresponding to Pb⁰ and Pb²⁺. The latter is likely due to the presence of a PbO surface layer, which forms very rapidly on Pb upon brief exposure to air.²⁵²⁶ The Pb²⁺ peak was almost completely removed by Ar⁺ sputtering (Figure S2), as seen in previous studies of Pb surfaces with a surface oxide. Measurements of the double layer charging currents in 0.5 M NaClO₄ indicated that the roughness factor for OD−Pb compared to Pb foil was 25 (Figure S3). Similar roughness factors were obtained for OD−Pb electrodes prepared in different electrolytes.

The H₂ evolution activities of OD−Pb and Pb foil were compared in constant-potential electrolyses performed under N₂ in three different electrolytes: 0.1 M KOH (pH 13), 0.25 M Na₂CO₃ (pH 11.5), and 0.325 M H₂SO₄/NaHSO₄ (pH 1.4). OD−Pb was formed from reduction of a thick PbO₂ layer in the electrolyte used for H₂ evolution electrolysis. The steady-state H₂ evolution current densities (j_H₂) were measured using separate electrolyses or a single stepped-potential electrolysis (see Supporting Information). Figure 2 shows Tafel plots of j_H₂, including the geometric j_H₂ for both electrodes and the surface-area-normalized j_H₂ for OD−Pb. H₂ evolution was suppressed on OD−Pb relative to Pb foil in all cases. Very large differences were observed in Na₂CO₃ and KOH, where the normalized j_H₂ on OD−Pb was up to ∼730-fold and ∼140-fold lower, respectively. A much smaller suppression was observed under acidic conditions, where normalized j_H₂ was only ∼8-fold lower on OD−Pb. The H₂ Tafel slopes were ∼200 mV dec⁻¹ in the acidic electrolyte and >250 mV dec⁻¹ in the alkaline electrolytes. The large slopes are likely caused by the presence of a Pb²⁺ surface layer that blocks H⁺ reduction (see the Discussion).

To see if the electrolyte in which the PbO₂ precursor is reduced affects the H₂ evolution activity, an OD−Pb electrode was prepared in 0.325 M H₂SO₄/NaHSO₄ and then evaluated in 0.25 M Na₂CO₃. The current versus time trace in Na₂CO₃ solution showed a brief initial period of relatively high current before reaching a steady-state value that was very close to the value obtained for an electrode prepared in Na₂CO₃ (Figure S4). The very large suppression of H₂ evolution on OD−Pb in Na₂CO₃ therefore does not require reducing the PbO₂ precursor in this electrolyte.

The CO₂ reduction activities of OD−Pb and polycrystalline Pb were compared in constant-potential electrolyses performed in CO₂-saturated 0.5 M NaHCO₃, pH 7.2. The amount of HCO₃⁻ produced was quantified by NMR analysis of the electrolyte. Gas chromatography of representative electrolyses indicated that H₂ was the only other significant product. Figure 3a shows the FE for HCO₃⁻ on both electrodes over a range of potentials for which appreciable CO₂ reduction was observed. On OD−Pb, the FE for HCO₃⁻ was ∼100% from −1.0 to −0.75 V and still ∼90% at −0.70 V. Nearly quantitative FE for HCO₃⁻ was maintained in a prolonged electrolysis of 75 h at −0.8 V (Figure S5). In contrast, Pb foil exhibited 50% to 60% FE for HCO₃⁻ from −1.0 to −0.85 V and only 30% at −0.8 V. The Tafel plots of the steady-state current density for HCO₃⁻ production (j_HCO₃⁻) are shown in Figure 3b. When corrected for the difference in surface areas, j_HCO₃⁻ was similar for OD−Pb and Pb foil from −0.8 to −1.0 V. The origin of the selectivity difference between OD−Pb and Pb foil is therefore the large suppression of H₂ evolution activity on OD−Pb.
These results demonstrate that it is possible to deactivate the H₂ evolution pathway on Pb while leaving CO₂ reduction unperturbed.

The Tafel slopes in the high overpotential regime were 164 mV dec⁻¹ on Pb foil and 195 mV dec⁻¹ on OD−Pb (Figure 3b). The reaction was approximately zeroth order in HCO₃⁻ concentration on OD−Pb at −0.95 V (Figure S6). The Tafel slope decreased to 83 mV dec⁻¹ from −0.7 to −0.78 V on OD−Pb, suggesting that there is a different mechanism in this potential range. The relatively low geometric current density for Pb foil precluded reliable measurement of jₑ for potentials > −0.8 V.

The selectivity difference between OD−Pb and Pb foil was magnified when electrolysis was performed in N₂-saturated 0.5 M NaHCO₃. Although there is very little equilibrium concentration of CO₂ under these conditions, dissociation of HCO₃⁻ near the electrode surface (2HCO₃⁻ ⇌ CO₂ + CO₃²⁻ + H₂O) supplies enough CO₂ to sustain some CO₂ reduction current density. At −1.0 V in N₂-saturated HCO₃⁻ solution, OD−Pb still had nearly quantitative FE for HCO₂⁻ while the FE on Pb foil was <10% (Figure 4). The total geometric current density was significantly higher on Pb foil than OD−Pb despite the higher surface area for the latter. Thus, the suppression of H₂ evolution on OD−Pb enables high selectivity for CO₂ reduction even in electrolytes with low CO₂ concentration.

**DISCUSSION**

The key functional difference between OD−Pb and Pb foil is the suppression of H₂ evolution on OD−Pb. The Tafel plots provide insight into the origin of this phenomenon. The Tafel slopes for both Pb foil and OD−Pb are very high in KOH and
Na$_2$CO$_3$ electrolytes, approaching a regime in which the current density is potential-independent. A previous study of H$_2$ evolution on Pb foil in H$_2$SO$_4$ and NaOH electrolytes also noted a propensity for high Tafel slopes. The slopes were lowered to 120 mV dec$^{-1}$ by using electrolytes that had been pre-electrolyzed for many hours and chemically polishing the Pb surface. The high Tafel slopes on Pb foil and OD−Pb in Figure 2 suggest that the surfaces are mostly passivated for H$^+$ reduction. Passivation may result from adsorption of electrolyte impurities, or the presence of a thin film on the surface.

The suppression of H$_2$ evolution from adsorption of electrolyte impurities would affect Pb foil much more than OD−Pb because of the lower surface area for Pb foil and therefore cannot explain the difference in their activity. We propose that there is a very thin, kinetically stable Pb oxide (hydroxide) layer on the surfaces of the electrodes that blocks H$_2$ evolution in KOH and Na$_2$CO$_3$ electrolytes. H$_2$ evolution takes place primarily at the remaining exposed Pb. The difference in activity between Pb foil and OD−Pb arises from a difference in the coverage of the passivating layer: the oxide covers a large portion (e.g., 90%) of the electrode in the case of Pb foil but an even larger portion (e.g., >99%) in the case of OD−Pb. In acidic electrolyte, the oxide layer is less stable and the coverage is much lower, which results in lower Tafel slopes and a much smaller difference in normalized activity between the two electrodes.

Despite the large difference in their H$_2$ evolution activity, OD−Pb and Pb foil have similar specific CO$_2$ reduction activity in HCO$_3^-$ electrolyte. This discrepancy implies that the Pb oxide (hydroxide) layer that passivates the electrodes for H$_2$ evolution is the active surface for CO$_2$ reduction. Because the coverage of this layer is high on both Pb foil and OD−Pb, the difference in the coverage between the two electrodes (e.g., 90% vs >99%) has a relatively small effect on the difference in their specific CO$_2$ reduction activity. In this respect, Pb is similar to Sn, which requires a Sn oxide layer for CO$_2$ reduction.

The Tafel plots and zeroth order dependence on HCO$_3^-$ for OD−Pb (Figures 3 and S6) are consistent with a CO$_2$ reduction mechanism in which the rate-determining step is the initial e$^-$ transfer to form a surface-bound CO$_2$$^*$. Tafel slopes of 120 mV dec$^{-1}$ are more typically observed for rate-limiting e$^-$ transfers, but the surface complexity of Pb under CO$_2$ reduction conditions could readily account for the higher slopes observed here.

The difference in coverage of a metastable Pb$^{2+}$ layer during catalysis on OD−Pb compared to Pb foil may arise from the relatively high density of defects on OD−Pb. In particular, the grain boundaries between the Pb nanocrystallites in OD−Pb could nucleate Pb$^{2+}$ oxide/hydroxide formation and thereby promote (nearly) full coverage of the electrode surface. This model implies that nanocrystalline or defect-rich Pb electrodes prepared by other methods would also show suppressed H$_2$ evolution activity, which may explain the transiently improved FE for CO$_2$ reduction on electrodeposited Pb observed previously. The ability of OD−Pb to maintain selective CO$_2$ reduction over long electrolyses may reflect greater stability of its morphology and microstructure compared to electrodeposited Pb.

**CONCLUSIONS**

In summary, OD−Pb has nearly quantitative FE for HCO$_3^-$ in both CO$_2$- and N$_2$-saturated HCO$_3^-$ solution at a range of overpotentials where the FE on Pb foil is only 5%–60%. The difference in selectivity arises because the normalized H$^+$ reduction activity is up to 700-fold lower on OD−Pb, but the intrinsic activity for CO$_2$ reduction to HCO$_3^-$ is not compromised. Based on the Tafel behavior, both electrodes appear to have a surface layer in alkaline electrolytes that blocks H$^+$ reduction but catalyzes CO$_2$ reduction at a low rate. This layer is likely to be a thin, metastable Pb$^{2+}$ oxide or hydroxide. H$^+$ reduction is suppressed on OD−Pb compared to Pb foil because the coverage of this layer is much higher on OD−Pb as a consequence of its microstructure and morphology.

**ASSOCIATED CONTENT**

Supporting Information

The following file is available free of charge on the ACS Publications website at DOI: 10.1021/cs5017672.

Experimental procedures and additional data (PDF)

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Notes

The authors declare no competing financial interest.

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